

Manganese antimony diselenide  
iodide,  $\text{MnSbSe}_2\text{I}$ Olivier Tougait,<sup>a\*</sup> James A. Ibers<sup>a</sup> and Arthur Mar<sup>b</sup><sup>a</sup>Department of Chemistry, Northwestern University, 2145 Sheridan Road, Evanston, IL 60208-3113, USA, and <sup>b</sup>Department of Chemistry, University of Alberta, Edmonton, AB, Canada T6G 2G2

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$\text{MnSbSe}_2\text{I}$  contains layers parallel to the  $bc$  plane that consist of edge- and corner-sharing  $\text{MnSe}_{6/2}$  and  $\text{MnSe}_{2/2}\text{I}_{4/2}$  octahedra. Sb atoms are located between these layers and form  $\text{SbSe}_3$  trigonal pyramids. Owing to the off-center placement of the Sb atom and the inequivalence of the two crystallographically independent Mn atoms, the monoclinic  $\text{MnSbSe}_2\text{I}$  structure is a distorted variant of the orthorhombic  $\text{UFeS}_3$  structure type.

## Comment

Quaternary phases consisting of a transition metal and several types of  $p$ -block elements in an ordered arrangement are relatively scarce. In  $\text{TMPnQX}$  systems (where TM is a transition metal;  $Pn = \text{P, As, Sb or Bi}$ ;  $Q = \text{S, Se or Te}$ ; and  $X = \text{F, Cl, Br or I}$ ), the most prevalent examples contain Cu, for example,  $\text{Cu}_6\text{PS}_5\text{Br}$  (Haznar *et al.*, 1999),  $(\text{CuBr})_3\text{P}_4\text{Se}_4$  (Reiser *et al.*, 2003),  $(\text{CuI})\text{P}_4\text{Se}_4$  (Pfitzner *et al.*, 1999),  $(\text{CuI})_3\text{P}_4\text{Se}_4$  (Pfitzner & Reiser, 1999),  $(\text{CuI})_2\text{P}_8\text{Se}_3$  (Pfitzner *et al.*, 2000),  $(\text{CuI})_2\text{Cu}_3\text{SbS}_3$  (Pfitzner, 1997),  $\text{Cu}_3\text{Bi}_2\text{S}_4\text{Cl}$  (Lewis & Kupčik, 1974) and  $\text{Cu}_3\text{Bi}_2\text{S}_4\text{Br}$  (Mariolacos & Kupčik, 1975), or another late  $d$ -block element, for example,  $\text{CdSb}_6\text{S}_8\text{I}_4$  (Sirota *et al.*, 1976) and  $\text{Hg}_3\text{AsQ}_4\text{X}$  ( $Q = \text{S or Se}$ , and  $X = \text{Cl, Br or I}$ ; Beck *et al.*, 2000). In most cases, the metal adopts a tetrahedral coordination, as expected for these late  $d$ -block elements; an interesting exception is  $\text{CdSb}_6\text{S}_8\text{I}_4$ , in which the Cd atom adopts an octahedral coordination. There appear to be no examples of compounds of formula  $\text{TMPnQ}_2\text{X}$ .

$\text{MnSbSe}_2\text{I}$  represents a new layered structure type, as shown in Fig. 1(a). The layers are parallel to the  $bc$  plane and contain two kinds of Mn-centered octahedra: atom Mn1 is surrounded by four Se2 and two Se1 atoms, whereas atom Mn2 is surrounded by four I1 and two Se1 atoms. The octahedra share their corners along the  $c$  direction and share their edges along the  $b$  direction. The Mn–Se distances of 2.6590 (9), 2.7849 (6) and 2.7687 (9) Å, and the Mn–I distances of 2.8380 (5) Å

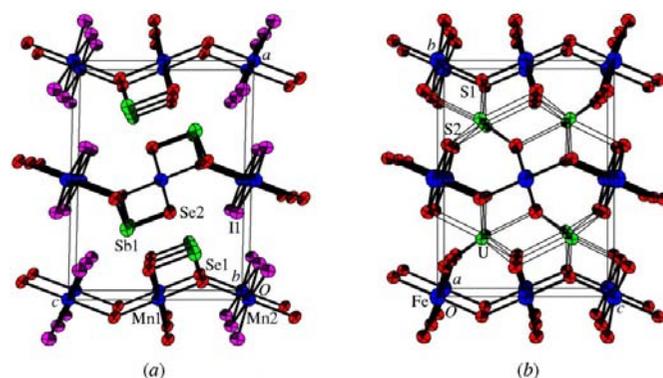


Figure 1

(a)  $\text{MnSbSe}_2\text{I}$  viewed along the  $b$  axis and (b)  $\text{UFeS}_3$  viewed along the  $a$  axis. Displacement ellipsoids are drawn at the 99% probability level in both cases.

(Table 1) are comparable to those found in olivine-type  $\text{Mn}_2\text{SiSe}_4$  [2.671 (4)–2.756 (3) Å; Jobic *et al.*, 1995] and in  $\text{CsMnI}_3$  [2.920 (2) Å; Zandbergen, 1980], respectively, where octahedrally coordinated Mn is present. Between the layers lie the Sb1 atoms, which are coordinated to three Se atoms, forming a trigonal pyramid, with Sb–Se distances [2.5995 (9) and 2.7277 (6) Å] that are similar to those in  $\text{Sb}_2\text{Se}_3$  [2.589 (1)–2.803 (1) Å; Voutsas *et al.*, 1985]. The closest distance between an Sb1 atom in one layer and an Se atom in the adjacent layer is 3.1523 (8) Å, which is too long to be considered as a covalent bond.

The structure of monoclinic  $\text{MnSbSe}_2\text{I}$  (space group  $C2/m$ ) is a distorted variant of orthorhombic  $\text{UFeS}_3$  ( $Cmcm$ ; Fig. 1b; Noël & Padiou, 1976). In the latter structure, layers of Fe-centered octahedra are also evident, but the intervening U atoms are arranged more symmetrically, residing in tricapped trigonal-prismatic sites. The lower monoclinic symmetry ( $\beta = 91.27^\circ$ ) of  $\text{MnSbSe}_2\text{I}$  arises from the absence of mirror and glide planes associated with the displacement of the Sb atoms, as well as the chemical inequivalence of the two types of Mn-centered octahedra. The irregular position of the Sb atom is indicative of  $\text{Sb}^{3+}$ , implying the presence of a lone pair. This assignment is consistent with the charge-balanced formulation  $(\text{Mn}^{2+})(\text{Sb}^{3+})(\text{Se}^{2-})_2(\text{I}^-)$ . Comparison with the analogous formulation  $(\text{U}^{3+})(\text{Fe}^{3+})(\text{S}^{2-})_3$  shows that replacement of one of the chalcogen atoms in  $\text{UFeS}_3$  by a halogen atom in  $\text{MnSbSe}_2\text{I}$  is compensated by substitution with a transition-metal ion of lower charge.

## Experimental

Single crystals of  $\text{MnSbSe}_2\text{I}$  were obtained as a by-product of the reaction of Sm (1.20 mmol, Alfa, 40 mesh, 99.9%), Mn (0.20 mmol, Alfa, 100 mesh, 99.9%), Sb (0.20 mmol, Alfa, 325 mesh, 99.5%) and Se (2.40 mmol, Alfa, 325 mesh, 99.99%) in a fused-silica tube containing a small amount of  $\text{I}_2$ . The reaction tube was heated at 1223 K for 4 d before being cooled to 973 K at a rate of 3 K  $\text{h}^{-1}$ , at which point the furnace was turned off. The product consisted mainly of golden plate-like crystals of  $\text{SmSe}_3$  and a few black prisms of  $\text{MnSbSe}_2\text{I}$ . Qualitative energy-dispersive X-ray analysis performed

on several of these prisms revealed an average composition (atomic %) of 19 (2)% Mn, 22 (2)% Sb, 39 (2)% Se and 21 (2)% I, in good agreement with that deduced from the single-crystal X-ray analysis. No trace of Sm was detected.

## Crystal data

MnSbSe <sub>2</sub> I	$D_x = 5.645 \text{ Mg m}^{-3}$
$M_r = 461.51$	Mo $K\alpha$ radiation
Monoclinic, $C_2/m$	Cell parameters from 2097 reflections
$a = 13.319 (3) \text{ \AA}$	$\theta = 3.1\text{--}28.6^\circ$
$b = 4.0359 (8) \text{ \AA}$	$\mu = 26.21 \text{ mm}^{-1}$
$c = 10.105 (2) \text{ \AA}$	$T = 153 (2) \text{ K}$
$\beta = 91.27 (3)^\circ$	Prism, black
$V = 543.08 (19) \text{ \AA}^3$	$0.19 \times 0.04 \times 0.02 \text{ mm}$
$Z = 4$	

## Data collection

Bruker SMART 1000 CCD area-detector diffractometer	737 independent reflections
$\omega$ scans	709 reflections with $I > 2\sigma(I)$
Absorption correction: numerical (XP in <i>SHELXTL</i> ; Sheldrick, 1997)	$R_{\text{int}} = 0.026$
$T_{\text{min}} = 0.145$ , $T_{\text{max}} = 0.656$	$\theta_{\text{max}} = 28.6^\circ$
2404 measured reflections	$h = -17 \rightarrow 17$
	$k = -5 \rightarrow 5$
	$l = -13 \rightarrow 13$

## Refinement

Refinement on $F^2$	$w = 1/[\sigma^2(F_o^2) + (0.04F_o^2)^2]$
$R(F) = 0.026$	$(\Delta/\sigma)_{\text{max}} < 0.001$
$wR(F^2) = 0.071$	$\Delta\rho_{\text{max}} = 2.11 \text{ e \AA}^{-3}$
$S = 1.37$	$\Delta\rho_{\text{min}} = -1.63 \text{ e \AA}^{-3}$
737 reflections	Extinction correction: <i>SHELXL97</i>
34 parameters	Extinction coefficient: 0.0022 (3)

**Table 1**

Selected interatomic distances ( $\text{\AA}$ ).

Sb1—Se2	2.5995 (9)	Mn1—Se2 <sup>iii</sup>	2.7849 (6)
Sb1—Se1 <sup>i</sup>	2.7277 (6)	Mn2—Se1 <sup>iv</sup>	2.7687 (9)
Mn1—Se1 <sup>ii</sup>	2.6590 (9)	Mn2—I <sup>iii</sup>	2.8380 (5)

Symmetry codes: (i)  $\frac{1}{2} - x, \frac{1}{2} - y, 1 - z$ ; (ii)  $-x, -y, 1 - z$ ; (iii)  $x - \frac{1}{2}, y - \frac{1}{2}, z$ ; (iv)  $-x, -y, -z$ .

Data collection: *SMART* (Bruker, 1999); cell refinement: *SAINT* (Bruker, 1999); data reduction: *SAINT*; program(s) used to solve structure: *SHELXTL* (Sheldrick, 1997); program(s) used to refine structure: *SHELXTL*; molecular graphics: *ATOMS* (Dowty, 1999); software used to prepare material for publication: *SHELXTL*.

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Supplementary data for this paper are available from the IUCr electronic archives (Reference: IZ1033). Services for accessing these data are described at the back of the journal.

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